Annealing effects on optical and structural properties of chromium oxide thin film deposited by PLD technique

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Abstract

Optical properties of chromium oxide (Cr$_2$O$_3$) thin films which were prepared by pulse laser deposition method, onto glass substrates. Different laser energy (500-900) mJ were used to obtain Cr$_2$O$_3$ thin films with thickness ranging from 177.3 to 372.4 nm were measured using Tolansky method. Then films were annealed at temperature equal to 300 °C. Absorption spectra were used to determine the absorption coefficient of the films, and the effects of the annealing temperature on the absorption coefficient were investigated. The absorption edge shifted to red range of wavelength, and the optical constants of Cr$_2$O$_3$ films increases as the annealing temperature increased to 300 °C. X-ray diffraction (XRD) study reveals that Cr$_2$O$_3$ thin films are amorphous; while the crystal structure of annealed Cr$_2$O$_3$ films is rhombohedral after annealing at 300 °C for two hour. AFM studies of Cr$_2$O$_3$ thin films exhibit a smooth and well dispersed on the surface.

Introduction

Chromium oxide (Cr$_2$O$_3$) nanoparticles are of interest for applications such as green pigments, wear resistance, thermal protection, digital recording systems, and chemical catalysts [1]. Cr$_2$O$_3$ nanoparticles have been prepared and characterized by many different researchers [2], and the resulting particle size depending on the method of preparation.

Chromium oxides (Cr$_2$O$_3$) are important technological materials because of their intrinsic properties, such as high temperature stability. In particular, among the many chromium oxides, Cr$_2$O$_3$ is the most
thermodynamically stable phase above 500°C, and is particularly interesting for its very high hardness and chemical resistance properties. It is used as a protective coating on high-temperature battery electrodes, as intermediate layers in corrosion-resistant applications on semiconductors and metals, as wear-resistant layers on magnetic particles, and as anti-oxidation layers on metals such as stainless steel [3, 4]. Liquid crystal displays used low-reflective Cr₂O₃/Cr films were used as black matrix films [5]. Literature reports indicate that Cr₂O₃ thin films have been produced by a number of techniques by a number of researchers. These include the vacuum evaporation [6], sputtering [4], Chemical Vapor Deposition (CVD) [5], Spray pyrolysis [7] and reactive pulsed laser ablation techniques [8].

This work contains the Structural, morphological, and optical properties of Cr₂O₃ thin films deposited by PLD at R.T., and then annealing at 300°C.

**Experimental procedure**

PLD technique is used to deposit Cr₂O₃ thin films shows in Fig. 1, where this technique consist of vacuum chamber evacuated to pressure (10⁻² mbar) and Q-switched Nd:YAG laser. The focused Nd:YAG laser beam at (500-900) mJ comes through a window and incident on surface of the Cr₂O₃ target with an angle of 45°. The substrate is placed in front of target and its surface parallel to it. Suitable gap is kept between substrate and target so that the substrate holder doesn't impede the incident laser beam. The films were deposited on glass at R.T. Q-switched Nd: YAG laser with (1064 nm) wavelength, (pulse width 10 ns) repetition frequency, (10 Hz), for 300 laser pulse used as deposition conditions.

![Fig.1: Pulse laser deposition setup using Nd: YAG laser.](image)

**Results and discussion**

The optical transmittance spectra for Cr₂O₃ thin films prepared by PLD technique at R.T., and annealed at 300°C; using different energies are shown in Figs. 2 and 3.

It is clear from these figures that for each analyzed thin film it has been noticed that the transmittance decreased with increasing annealing temperature. This is probably due to the improvement in the size of the crystal. It is clear that the annealing temperature has a clear impact on the transmittance decreasing. Transmittance decrease with increase temperature, the Cr₂O₃ films annealed at higher temperatures shows less transmittance values. In the Figs. 2 and 3, the greatest value of transmittance is
in the Near IR range of the spectrum (300-1100) nm almost 99 % at R.T. and 90 % at 300 °C annealing temperature for energy 500 mJ and wavelength equal to 1000 nm. While the lowest value of transmittance is 82 % at R.T. and 74 % for annealed film at energy 900 mJ and wavelength equal to 1000 nm which is well agreed with reported values [9]. Also the optical transmittance depends on the thickness of the film. As the thickness of the thin film increases the transmittance decreases i.e. inverse proportionality between them. It is noted in all Cr$_2$O$_3$ films prepared by PLD that optical transmittance value be the highest at high wavelengths. For annealing samples, it is observed that the overall maximum peak shifts towards the shorter wavelength with decrease the energy. This is because in the case of thicker films, more atoms are present in the film so more states will be available for the photons to be absorbed.

![Graph](image1)

*Fig. 2: Transmission versus wavelength of Cr$_2$O$_3$ films prepared by PLD at R.T. with different energies.*

![Graph](image2)

*Fig. 3: Transmission versus wavelength of Cr$_2$O$_3$ films prepared by PLD at 300 °C annealing temperature with different energies.*

We can see from Tables 1 and 2, Figs. 4 and 5, that the absorption coefficient of the Cr$_2$O$_3$ films is characterized by strong absorption at
the shorter wavelength region between 300-600 nm at R.T. and 200-350 nm for annealed films. In the shorter wavelength the absorption coefficient exhibits high values of $\alpha$ ($\alpha > 10^4$ cm$^{-1}$) which means that there is a large probability of the allowed direct transition [10], and then $\alpha$ decreases with increasing of wavelength. Also we can notice from these figures that $\alpha$ increase with increasing of energies of pulse as given in Table 1 and Fig. 4 for samples prepared at R.T. Values of $\alpha$ increase between (0.79–3.22) $\times 10^4$ cm$^{-1}$ with increasing of energies (500-900) mJ at $\lambda = 500$ nm wavelength. Same behavior shown in Fig. 5 and Table 2 for samples annealed at 300 °C the values of $\alpha$ increase between (0.60–3.26)$\times 10^4$ cm$^{-1}$ with increasing of energies from 500 to 900 mJ.

<table>
<thead>
<tr>
<th>$E$ (mJ)</th>
<th>T%</th>
<th>$\alpha$ (cm$^{-1}$)</th>
<th>$\kappa$</th>
<th>$n$</th>
<th>$\omega$</th>
<th>$\rho$</th>
<th>$E_g$ (eV)</th>
<th>$E_g$ (eV)</th>
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<tr>
<td>500</td>
<td>72.81</td>
<td>7934</td>
<td>0.032</td>
<td>2.690</td>
<td>6.970</td>
<td>0.167</td>
<td>2.7</td>
<td>3.59</td>
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<td>600</td>
<td>65.26</td>
<td>10560</td>
<td>0.042</td>
<td>3.058</td>
<td>9.227</td>
<td>0.258</td>
<td>2.55</td>
<td>3.5</td>
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<tr>
<td>700</td>
<td>58.30</td>
<td>13403</td>
<td>0.053</td>
<td>3.410</td>
<td>11.622</td>
<td>0.364</td>
<td>2.6</td>
<td>3.39</td>
</tr>
<tr>
<td>800</td>
<td>51.77</td>
<td>16460</td>
<td>0.066</td>
<td>3.791</td>
<td>14.368</td>
<td>0.497</td>
<td>2.5</td>
<td>3.30</td>
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<tr>
<td>900</td>
<td>27.49</td>
<td>32282</td>
<td>0.129</td>
<td>5.004</td>
<td>25.023</td>
<td>1.286</td>
<td>2.35</td>
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Table 1: Optical constants of Cr$_2$O$_3$ thin films prepared by PLD at R.T., and at $\lambda = 500$ nm.

Fig. 4: The absorption coefficient versus wavelength of Cr$_2$O$_3$ films coated glass prepared by PLD at R.T. with different energies.

Fig. 5: The absorption coefficients versus wavelength of Cr$_2$O$_3$ films prepared by PLD at annealed 300 °C with different energies.
Table 2: Optical constants of Cr$_2$O$_3$ thin films prepared by PLD annealed at 300 °C, and at $\lambda = 500$ nm.

<table>
<thead>
<tr>
<th>E (mJ)</th>
<th>T%</th>
<th>$\alpha$ (cm$^{-1}$)</th>
<th>k</th>
<th>n</th>
<th>$\varepsilon_r$</th>
<th>$\varepsilon_i$</th>
<th>Eg (eV)</th>
</tr>
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<tr>
<td>500</td>
<td>78.35</td>
<td>6098</td>
<td>0.024</td>
<td>2.358</td>
<td>5.558</td>
<td>0.114</td>
<td>4.05</td>
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<tr>
<td>600</td>
<td>73.72</td>
<td>7623</td>
<td>0.030</td>
<td>2.593</td>
<td>6.724</td>
<td>0.157</td>
<td>4.02</td>
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<tr>
<td>700</td>
<td>62.11</td>
<td>11907</td>
<td>0.047</td>
<td>3.209</td>
<td>10.298</td>
<td>0.304</td>
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<td>800</td>
<td>61.97</td>
<td>11964</td>
<td>0.048</td>
<td>3.217</td>
<td>10.349</td>
<td>0.306</td>
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<td>900</td>
<td>27.13</td>
<td>32616</td>
<td>0.130</td>
<td>5.014</td>
<td>25.124</td>
<td>1.302</td>
<td>3.80</td>
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$E_g$ of Cr$_2$O$_3$ thin films prepared by PLD annealed at 300 °C larger than $E_g$ of Cr$_2$O$_3$ thin films prepared by PLD at R.T., due to increasing the temperature, the energy gap increases because the atoms are re-positioned to the correct locations and energy levels disappears accordingly, then, increasing the energy gap.

Fig. 6 shows the XRD results of Cr$_2$O$_3$ films deposited at R.T. by PLD for different energies (500-900) mJ and annealed at 300 °C to form phase and achieve the recrystallization process and to get large Cr$_2$O$_3$ particles. We deduce that our samples are effectively contains polycrystalline structure. In preference, diffraction peaks at crystalline planes (112), (104), (110), (113), (024), (116) and (300). The peak positions of the Cr$_2$O$_3$ planes were observed at 24°, 33.5°, 36°, 41°, 50.5°, 55° and 66°. All the observed peaks are in well agreement with the reported results [11].

All the diffraction peaks in the XRD pattern can be indexed to phase of Cr$_2$O$_3$ films deposited by PLD on glass substrates annealed at 300 °C, which are in good agreement with the standard values of the reported data (JCPDS no. 38-1479) [12] as the major phase and no secondary phase with a preferred orientation along (104) direction.

![Fig. 6: X-ray diffraction pattern of Cr$_2$O$_3$ films prepared by PLD and annealed at 300 °C with different energies.](image-url)
Fig. 7 shows AFM images of the surface topography of Cr$_2$O$_3$ thin films. Surface topography parameters and average grain size which were estimated from the granularity copulation distribution are tabulated in Table 3 which is well agreed with reported values [13]. The images of Cr$_2$O$_3$ thin film prepared at R.T. shows the surface with uniform island-like topography and some structure of clusters can be clearly observed in figures. These crystallite clusters are the result of crystallites coalescence. The grain sizes of grains in clusters of Cr$_2$O$_3$ thin films are in the range of nanometers. In these images, Surface Roughness Average $S_a$ is small which shows very good smoothness of the surface. This means that the prepared films are well deposited.

Fig. 7: 3D AFM topography image and distribution chart of Cr$_2$O$_3$ film deposited by PLD at R.T. with different energies.
Table 3: Surface topography parameters and grain sizes of Cr$_2$O$_3$ thin films.

<table>
<thead>
<tr>
<th>Energy (mJ)</th>
<th>Thin film thickness (nm)</th>
<th>Ave. Roughness (nm)</th>
<th>Root Mean Sq. (nm)</th>
<th>Peak To Peak (nm)</th>
<th>Grain size (nm)</th>
</tr>
</thead>
<tbody>
<tr>
<td>500</td>
<td>177.3</td>
<td>4.32</td>
<td>5.9</td>
<td>19.2</td>
<td>80.60</td>
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<tr>
<td>600</td>
<td>199.5</td>
<td>0.375</td>
<td>0.443</td>
<td>1.7</td>
<td>82.64</td>
</tr>
<tr>
<td>700</td>
<td>210</td>
<td>1.23</td>
<td>1.22</td>
<td>4.72</td>
<td>86.62</td>
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<tr>
<td>800</td>
<td>249.3</td>
<td>2.19</td>
<td>2.59</td>
<td>9.81</td>
<td>97.92</td>
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<tr>
<td>900</td>
<td>372.4</td>
<td>3.43</td>
<td>4</td>
<td>14.9</td>
<td>102.90</td>
</tr>
</tbody>
</table>

Conclusions

The optical energy gap (E$_g$) of the Cr$_2$O$_3$ films decreases with increasing lasing energy and increased with increased annealing temperature. It is clear that the annealing temperature has a clear impact on the transmittance decreasing; this is a result of increasing the surface roughness of the thin film and increase the scattering of light. Also, the optical transmittance depends on the thickness of the film. The absorption coefficient decreases with increasing of wavelength and increase with increasing of energies.

References